# Competition between luminescence and nonradiative processes in wide-gap scintillation materials



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Wide-gap binary and complex metal oxides doped with rare earth or transition impurity ions (luminescence centers) – fast scintillators, spectral transformers.

The energy absorbed during irradiation is transformed into:

- luminescence
- *heat release* (phonon package)
- creation of radiation defects

(short-lived or stable interstitial-vacancy pairs of Frenkel defects)

<u>The ratio</u> between three transformation channels <u>depends on</u>:

- $\rightarrow$  temperature,
- → concentration and <u>spatial distribution</u> of <u>luminescence</u> centers (presence of pairs of heavy impurity ions!)
  - → density of radiation-induced electronic excitations.

Scintillator energetic yield ≤ 1/3, CsI:Tl ~12%

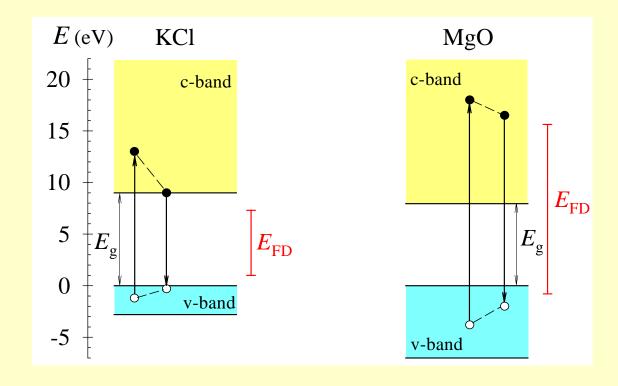
#### **Irradiation providing** *low* (e.g., X-rays)

or

#### superhigh excitation density

(~GeV Au<sup>197</sup> or U<sup>238</sup> ions: LET > 20 keV/nm, R~ 50–100 µm,  $>99.9\% \rightarrow ionisation losses)$ 

#### Materials with $E_{\rm FD} < E_{\rm g}$ , Materials with $E_{\rm FD} > E_{\rm g}$



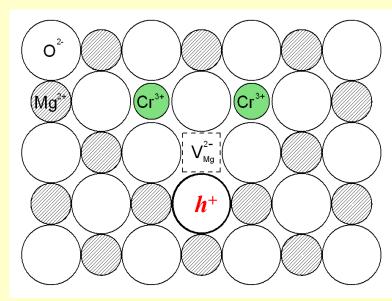
**Electronic excitations** via transitions from core bands!

 $E_{\rm FD}$ - creation energy of an *i-v* pair of Frenkel defects

- Pure (< 1 ppm of impurities) and stoichiometric MgO single crystals exhibit extremely high resistance against irradiation
- MgO doped with ~10 ppm of light Be<sup>2+</sup>, OH<sup>-</sup> sharp decrease of resistance
- MgO:Cr<sup>3+</sup> (~1000 ppm, heavy  $^{52}$ Cr- $^{52}$ Cr) [Cr<sup>3+</sup>- $v_c$ -Cr<sup>3+</sup>]<sup>0</sup>

Rhombic pair chromium center containing a cation vacancy

$$e + h[Cr^{3+}-v_c-Cr^{3+}]^0 \rightarrow defects$$



Rhombic center is a trap for a hole that becomes localized at oxygen. Coulomb center possesses a large cross-section for the recombination with electrons (hot conduction electrons or 100–300-eV electrons formed via transitions between inner shells).

Luminescence efficiency of  $[Al^{3+}-v_c-Al^{3+}]$  is close to 1 at direct excitation (even after X-irradiation),  $Al^{3+}$  is two times lighter than  $Cr^{3+}$ 

Theoretical predictions (A.J. Sievers, V. Hizhnyakov, et al.) — a new kind of discrete vibronic mode (<u>discrete breathers or discrete solitons</u>) with a large amplitude and frequency above the top of the acoustic phonon spectrum can be excited in a crystal lattice predisposed to anharmonic interactions.

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A.J. Sievers and S. Takedo, PRL <u>61</u>, 970 (1988).

M. Haas, V. Hizhnyakov, A. Shelkan, M. Klopov, A. J. Sievers, PRB <u>84</u>, 144303 (2011).

H. Guo, M. Zhang, J. Han, H. Zhang, N. Song, Physica B <u>404</u>, 2262 (2012).
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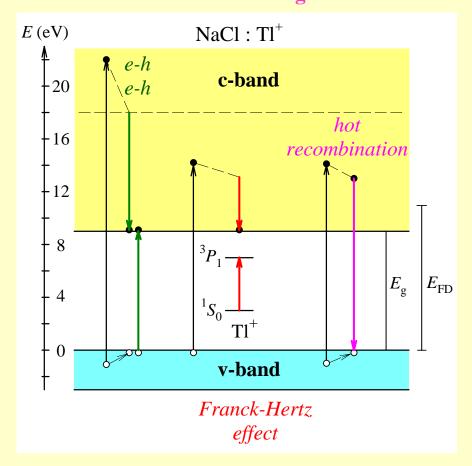
These unusual excitations are expected to be especially strong in wide-gap materials built up of ions with different masses

Under high-dense excitation (e.g., by swift heavy ions)

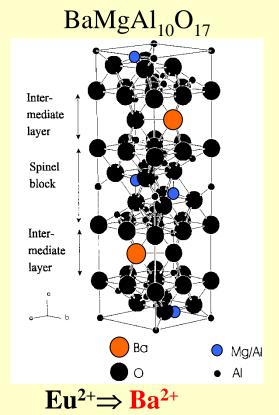
## collapse of discrete soliton $\rightarrow$ cooperative rearrangement of many host ions, creation of nanosize 3D defects

V.E. Zakharov and E.A. Kuznetsov, Phys. Usp. <u>55</u>, 535 (2012) [UFN <u>55</u>, 569 (2012)].

"Luminescent protection" against non-impact creation of Frenkel defect in wide-gap materials with  $E_{\rm FD} > E_{\rm g}$ .



A solid-state analogue of the Franck-Hertz effect - a hot conduction electron can spent its energy excess for the direct excitation of an impurity center, while a subsequent recombination of the cooled e with a valence h does not create a Frenkel pair.



10 25 30 20 5 15 Photon Energy (eV) Excitation spectra for the emission of Eu<sup>2+</sup> centers at 295 K and the 5.1 eV intrinsic emission at 8 K

 $Ba_{0.9}MgAl_{10}O_{17}:Eu_{0.1}^{2+}$ 

 $E^{\mathrm{e-h}}$ 

spatially separated impurity ions

Ba<sub>0.9</sub>Eu<sub>0.1</sub>MgAl<sub>10</sub>O<sub>17</sub> phosphors demonstrate high QY and reasonably high resistance against electron or VUV irradiation.

Quantum efficiency

In BaAl<sub>11</sub>O<sub>16</sub>N:Eu – MgO is replaced by AlN and Eu<sup>2+</sup> is located near 3 aluminum vacancies. As a result, both QY and radiation resistance are lower (similarity with MgO:Cr).

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Intensity

BaMgAl<sub>10</sub>O<sub>17</sub>

### $E_{\rm FD} > E_{\rm g}$

F<sup>+</sup> centers

Antisite defects

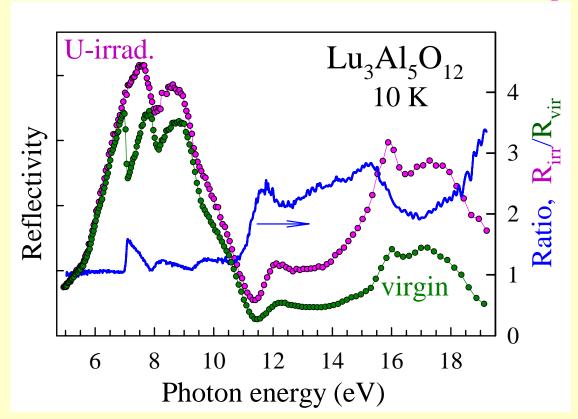
host

impurity

$$\text{Lu}^{3+}\Big|_{\text{Al}^{3+}} \text{ or } \text{Ce}^{3+}\Big|_{\text{Al}^{3+}}$$

Impurity pairs:

$$Ce^{3+}\Big|_{Lu^{3+}} - Ce^{3+}\Big|_{Al^{3+}}$$



Reflection spectra of LuAG single crystal before and after irradiation with  $^{238}$ U ions (2.14 GeV,  $10^{12}$  ions cm<sup>-2</sup>, RT) at 10 K. The ratio  $R_{irr}/R_{vir}$ .

place for the collapse of discrete breathers creation of temperature-stable 3D defects

#### **Concluding Remarks**

High light yield and acceptable radiation resistance can be simultaneously reached only in doped wide-gap materials where not only *valence* and *ion radius* but also an **atom mass** are close for all elements of a solid solution. CsI:Tl<sup>+</sup>, LaBr<sub>3</sub>:Ce<sup>3+</sup> and SrBr<sub>2</sub>:Eu<sup>2+</sup> fulfill all these requirements.

In LiI:Eu<sup>2+</sup>, LaCl<sub>3</sub>:Ce<sup>3+</sup>, CaSO:Tb<sup>3+</sup>,F<sup>-</sup> both LY and radiation resistance are low even at low impurity concentration.

Especially harmful is the presence of the <u>pairs of spatially close heavy rare-earth impurity ions</u> in cation sites.

Mass disbalance results in the appearance of *discrete breathers*, the collapse of which under high and superhigh excitation density takes place at pair-impurity centers and leads to the creation of complex defects. 3D defects serve as stoppers for dislocations and cause a cracking of single crystals.

It seems that if a pair consists of heavy but different impurity ions (for instance, Ce<sup>3+</sup> and Tb<sup>3+</sup>), the formation of a unified complex molecule is unlikely and a material is more resistant against radiation as compared to the presence of pairs of identical RE ions.



## THANK YOU