

Luminescent and radiation-induced properties of $M_{1-x}Pr_xF_{2+x}$ ($M^{2+}=Ca, Sr, Ba, x=0.35$) solid solutions

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New potential applications of fluorides

✓ Homogeneous Hadron Calorimeter Detector (HHCAL)

The problem: dual readout of Cherenkov and scintillation light to achieve good resolution for hadrons and jets

Requirements to Materials:

High density ($>7\text{g/cm}^3$) and Z_{eff}

wide E_g (good UV transmittance: UV cut-off < 350 nm)

Some scintillation light, *not necessary* bright and fast

Candidates: PbF_2 , ...?

✓ Dark matter search

The problem: detecting Weakly Interacting Massive Particles (WIMP)

Requirements to Materials:

Discrimination the weak and rare signal over the background caused by natural radioactivity

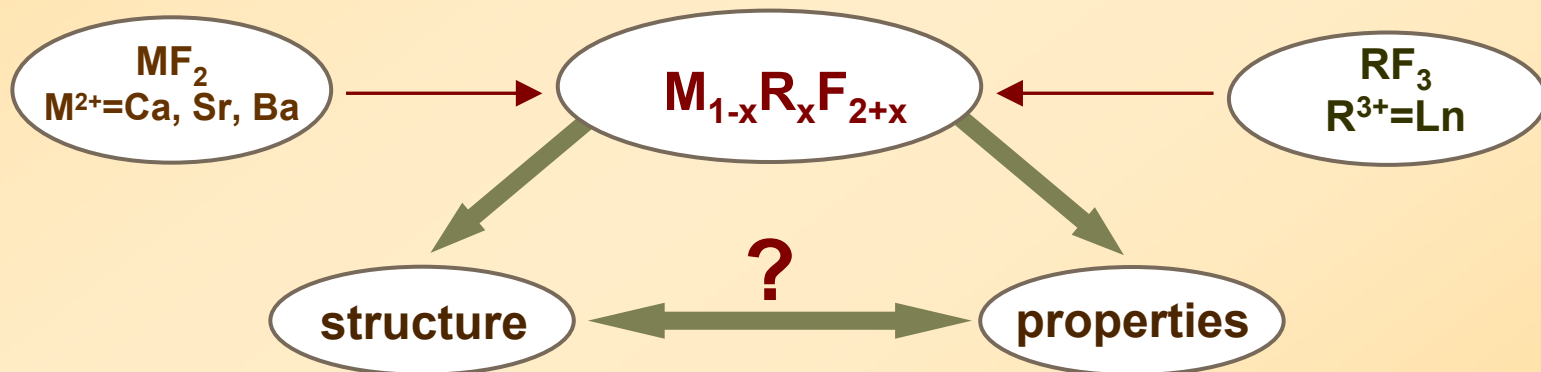
High interaction rates with WIMPs

Sufficient light yield at low temperature

Candidates: LiF , $\text{CaF}_2:\text{Eu}$, ...?

From the engineering point of view some attempts can be made to modify **luminescent properties** of fluorides via modification of **crystalline host**

In this case **mixed $M_{1-x}R_xF_{2+x}$ crystals** are of interest as model objects, since R^{3+} ions become the **part of matrix cations**

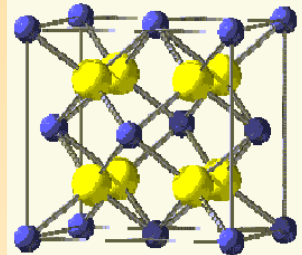


The main question:

How the modification of the crystal structure will influence on luminescent and radiation induced properties of the obtained mixed crystals?

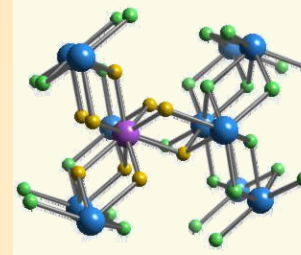
$M_{1-x}R_xF_{2+x}$ ($M^{2+}=\text{Ca, Sr, Ba}$, $R^{3+}=\text{Ln}$) – mixed crystals

MF_2 ($M^{2+}=\text{Ca, Sr, Ba}$)



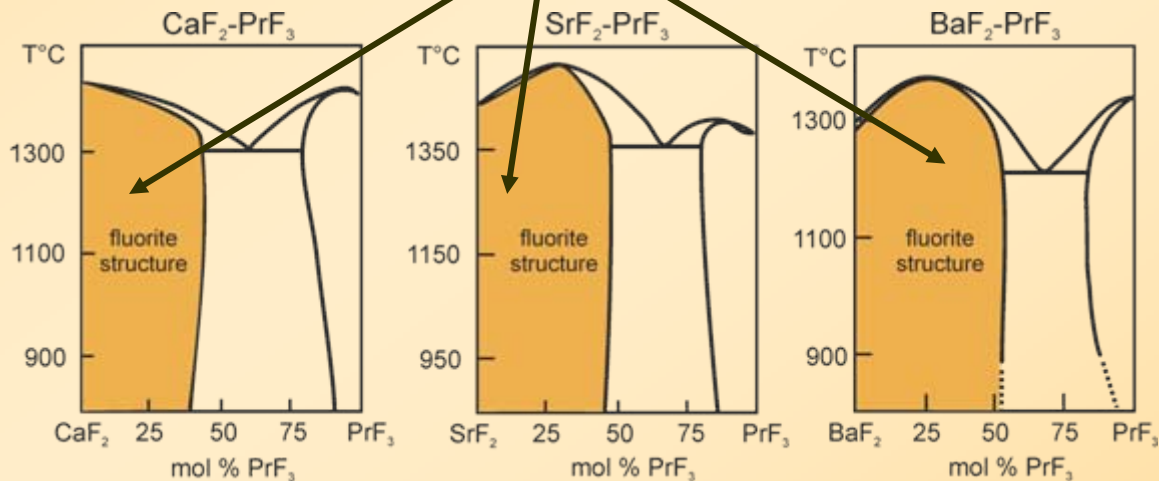
fluorite structure

RF_3 ($R^{3+}=\text{La-Nd}$)



tysonite structure

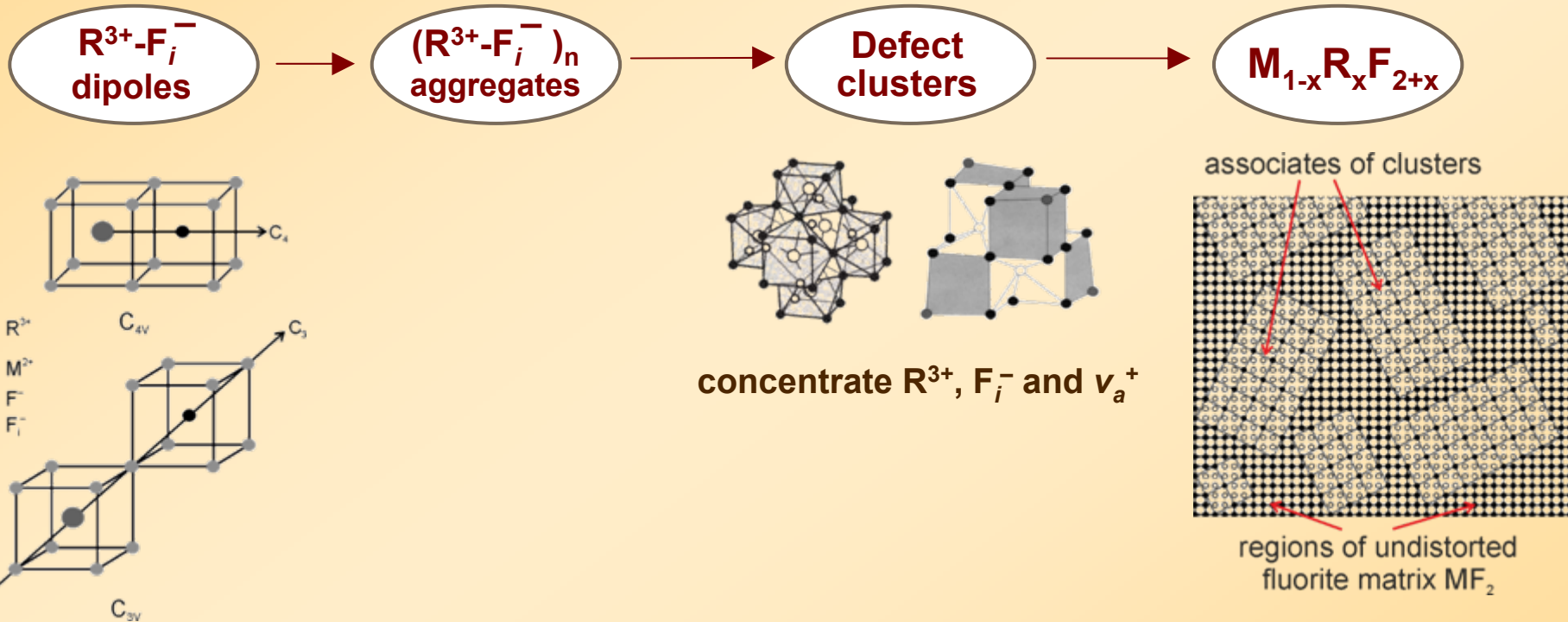
+



- ✓ high reciprocal solubility of MF_2 and RF_3 despite different structure types
- ✓ wide concentration range ($x \leq 0.5$) of fluorite-type $M_{1-x}R_xF_{2+x}$ solid solutions

Formation of $M_{1-x}R_xF_{2+x}$ solid solutions

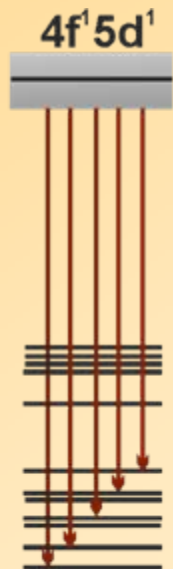
$C \leq 0.01 \text{ mol\%}$ $\xrightarrow{\text{RF}_3 \text{ concentration increase in the solid solution}}$ $C \sim 20\text{-}50 \text{ mol\%}$



B.P. Sobolev, 2003

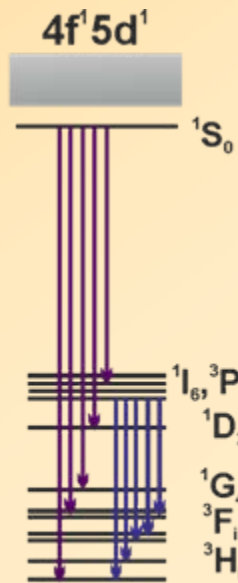
Modification of crystal structure via increasing of R^{3+} concentration

Pr^{3+}



$4f^1 5d^1 \rightarrow 4f^2$

coord. numb. ≤ 8



Photon Cascade Emission (PCE)

coord. numb. ≥ 9

Pr^{3+} is suitable as “probe” ion due to the strong dependence of luminescent properties on local environment

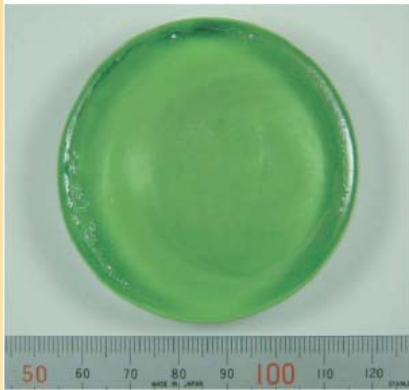
“Background”: appearance of PCE centers in MF_2 -Pr with increase in Pr^{3+} concentration [E.A.Radzhabov, 2009; P. A. Rodnyi 2005]

P. Rodnyi, 2002; Makhov, 2003

$MF_2:Pr$,
 $KYF_4:Pr$
 $LiKYF_5:Pr$

PrF_3
 $YF_3:Pr$
 $KMgF_3:Pr$
 $SrYF_5:Pr$

- ✓ **Characterization of $M_{1-x}Pr_xF_{2+x}$ crystals**
- ✓ **Different types of emission centers in $M_{1-x}Pr_xF_{2+x}$ solid solutions and their possible origin**
- ✓ **Radiation damage**
- ✓ **Conclusions**



$\text{Ca}_{0.65}\text{Pr}_{0.35}\text{F}_{2.35}$ crystal

$\text{M}_{1-x}\text{Pr}_x\text{F}_{2+x}$ crystals:

- ✓ grown by Bridgeman technique in CF_4 atmosphere
(Institute for Materials Science, Tsukuba, Japan)
- ✓ free from oxygen-containing impurities
- ✓ have the fluorite structure ($Fm3m$)
- ✓ no presence of PrF_3 phase

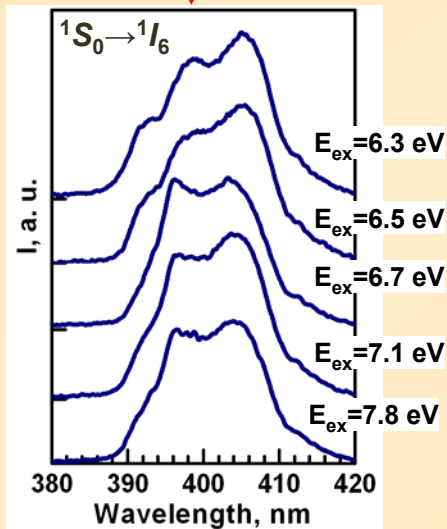
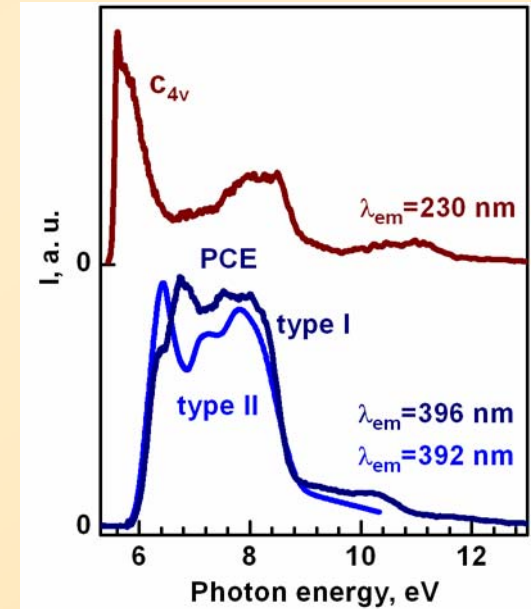
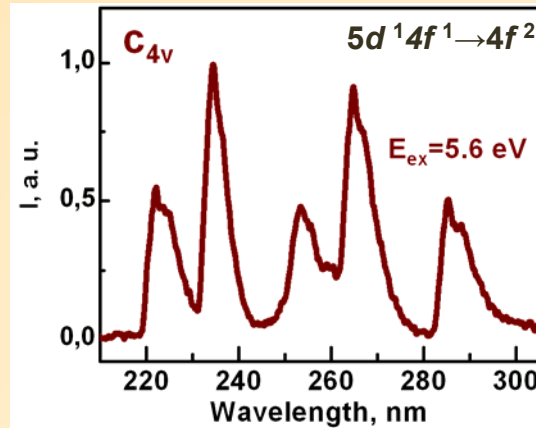
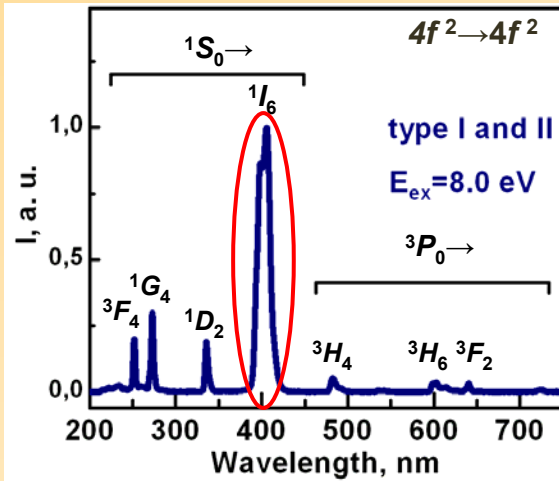
Crystal	Lattice constant, Å	Structure type
CaF_2	5.463	fluorite $Fm3m$
$\text{Ca}_{0.65}\text{Pr}_{0.35}\text{F}_{2.35}$	5.613	
SrF_2	5.799	
$\text{Sr}_{0.65}\text{Pr}_{0.35}\text{F}_{2.35}$	5.815	
BaF_2	6.200	
$\text{Ba}_{0.65}\text{Pr}_{0.35}\text{F}_{2.35}$	6.037	

Techniques

- ✓ luminescent spectroscopy under SR excitation (SUPERLUMI, DESY, Hamburg)
- ✓ absorption
- ✓ thermostimulated luminescence (TSL)

Emission

Photon Cascade Emission (PCE)

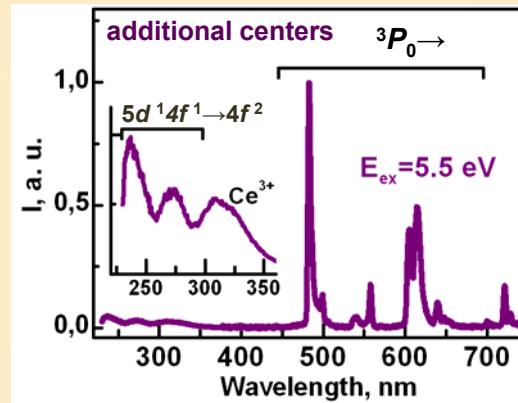
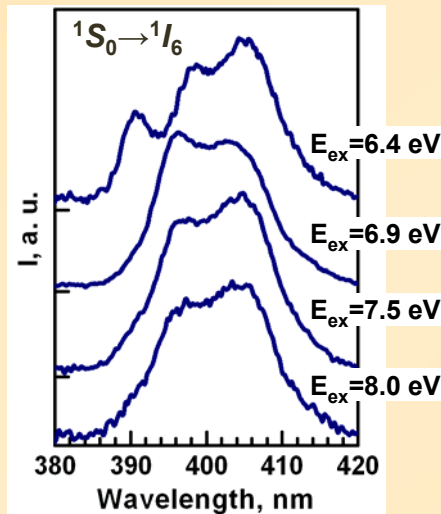
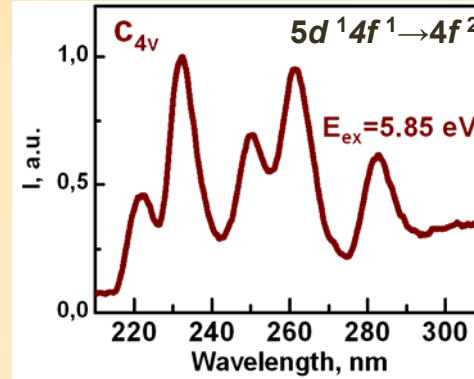
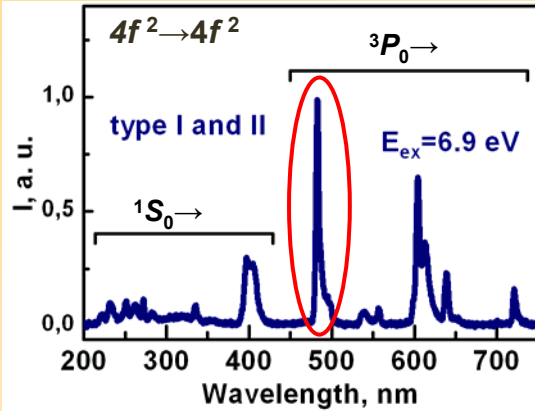


Different emission centers

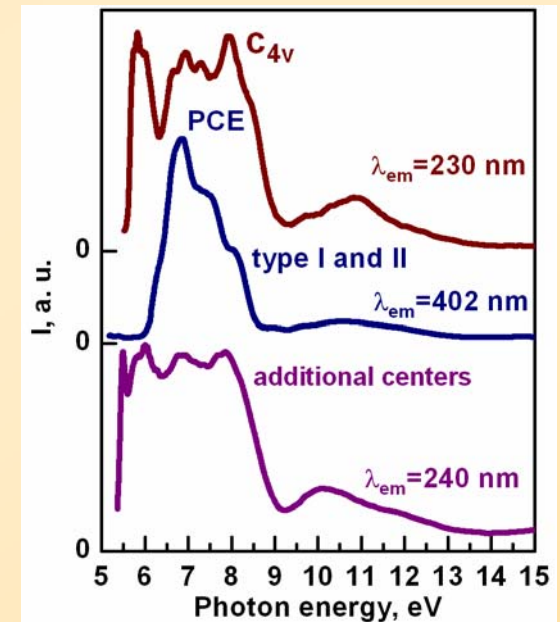
- ✓ PCE centers (two types) – dominant
- ✓ $5d^1 4f^1 \rightarrow 4f^2$ is typical for single Pr^{3+} ions in C_{4v} symmetry sites

Emission

Photon Cascade Emission (PCE)

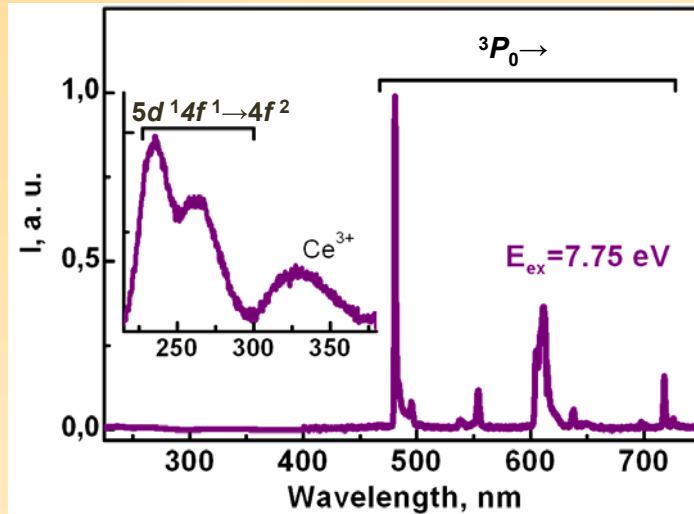


Excitation

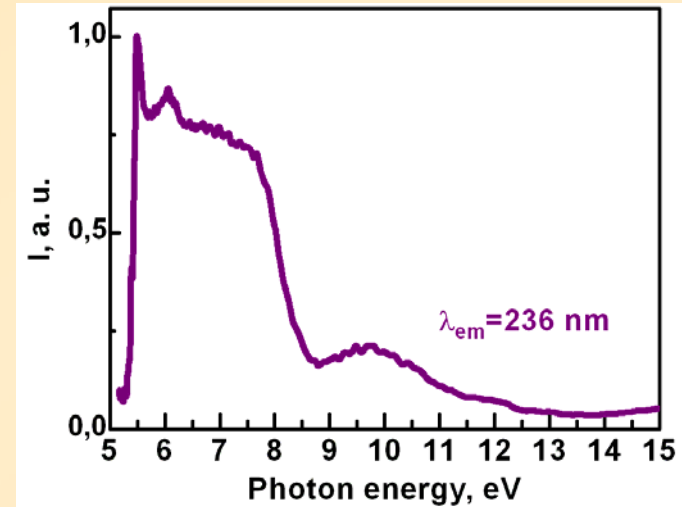


- ✓ single Pr^{3+} ions in C_{4v} sites
- ✓ two types of PCE centers
- ✓ additional type of centers (absent in $\text{Ca}_{0.65}\text{Pr}_{0.35}\text{F}_{2.35}$)

Emission



Excitation

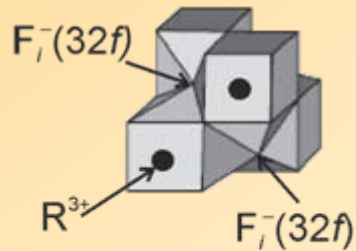


- ✓ only one type of emission center
- ✓ luminescent properties of observed centers in $\text{Ba}_{0.65}\text{Pr}_{0.35}\text{F}_{2.35}$ and additional centers in $\text{Sr}_{0.65}\text{Pr}_{0.35}\text{F}_{2.35}$ are similar
- ✓ low efficiency of $5d\ 14f\ 1 \rightarrow 4f\ 2$ emission alongside with intense transitions from 3P_0 level may be due to cross-relaxation between adjacent Pr^{3+} ions via the following scheme : $5d\ 14f\ 1 + {}^3H_4 \rightarrow {}^3P_0 + {}^3P_0$

Possible origin of emission centers in $M_{1-x}Pr_xF_{2+x}$

$$r(R^{3+})/r(M^{2+}) \sim 1.02-1.08$$

Clusters
1:0:4 (1:0:3)



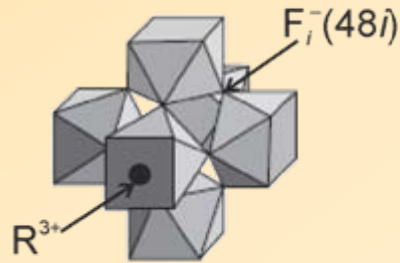
up to 3 R^{3+} ions

R^{3+} coord. numb. = 10(9)
irregular 10(9)-vertices
polyhedron

B.P. Sobolev, 2003; J.P. Laval, 1989I

$$r(R^{3+})/r(M^{2+}) \sim 0.7-0.8$$

Cluster
8:12:0



up to 6 R^{3+} ions

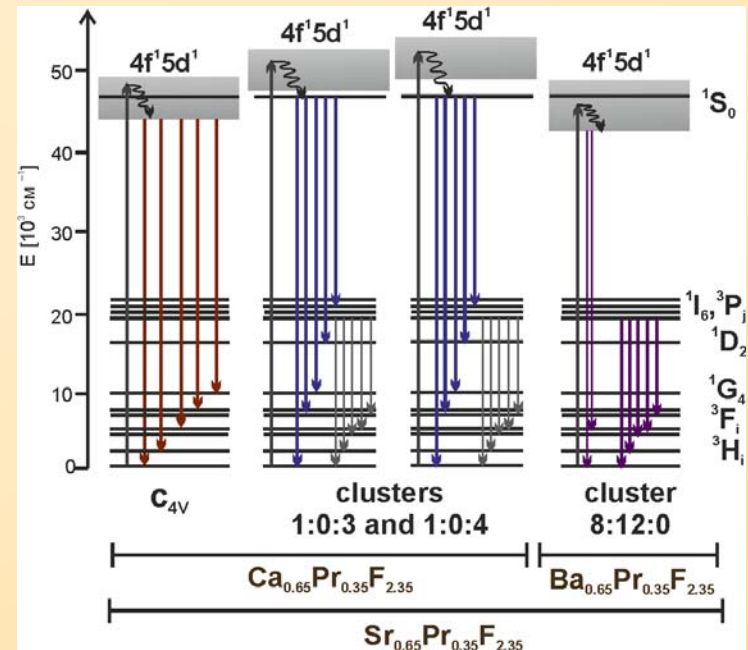
R^{3+} coord. numb. = 8
distorted square
antiprism

Crystal	$r(Pr^{3+}), \text{\AA}$	$r(M^{2+}), \text{\AA}$	$r(Pr^{3+}) / r(M^{2+})$
$Ca_{0.65}Pr_{0.35}F_{2.35}$	1.06	0.99	1.07
$Sr_{0.65}Pr_{0.35}F_{2.35}$		1.12	0.95
$Ba_{0.65}Pr_{0.35}F_{2.35}$		1.34	0.79

Crystal	(Pr^{3+}, free)	Possible origin of center					
		clusters					
		1:0:3		1:0:4		8:12:0	
E_{fd}, eV	E_{fd}, eV	D, eV	E_{fd}, eV	D, eV	E_{fd}, eV	D, eV	
$Ca_{0.65}Pr_{0.35}F_{2.35}$		6.46	1.12	6.90	0.68	-	
$Sr_{0.65}Pr_{0.35}F_{2.35}$	7.58	6.40	1.18	6.85	0.73	5.46	2.12
$Ba_{0.65}Pr_{0.35}F_{2.35}$		-		-		5.49	2.08

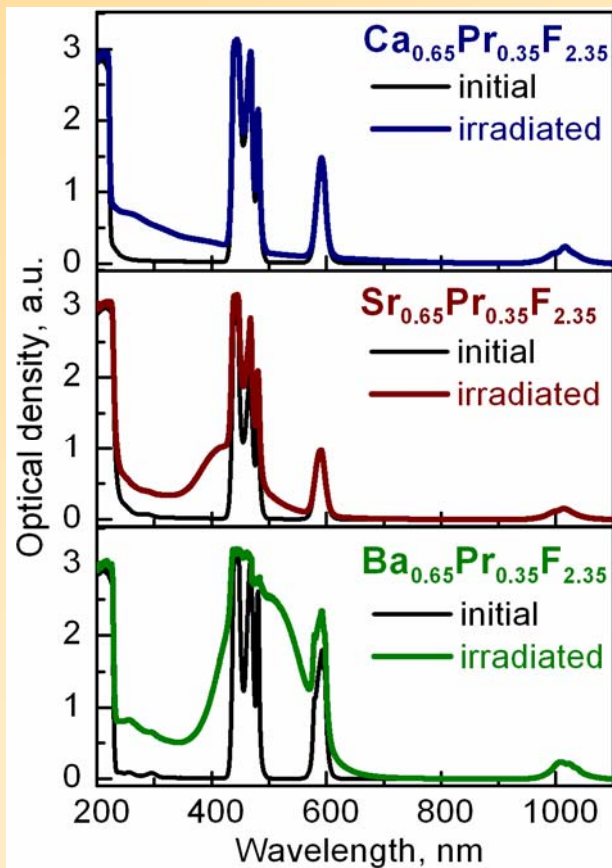
$$D(\text{"redshift"}) = E_{fd}(R^{3+}, \text{free}) - E_{fd}(R^{3+}, A)$$

[P. Dorenbos, 2003]



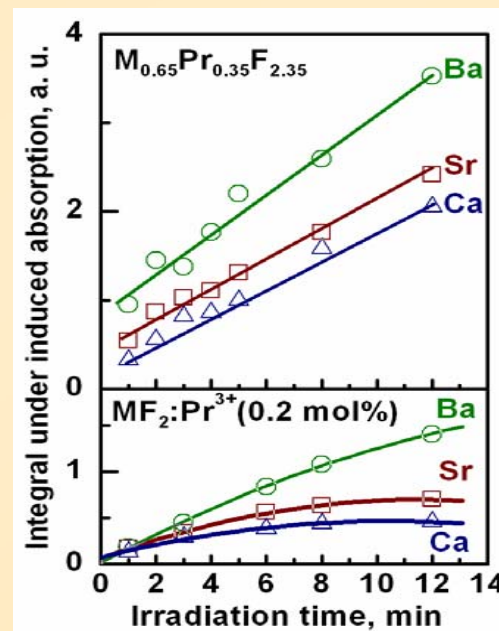
X-irradiation, 300 K

Absorption spectra, D=80 Gy



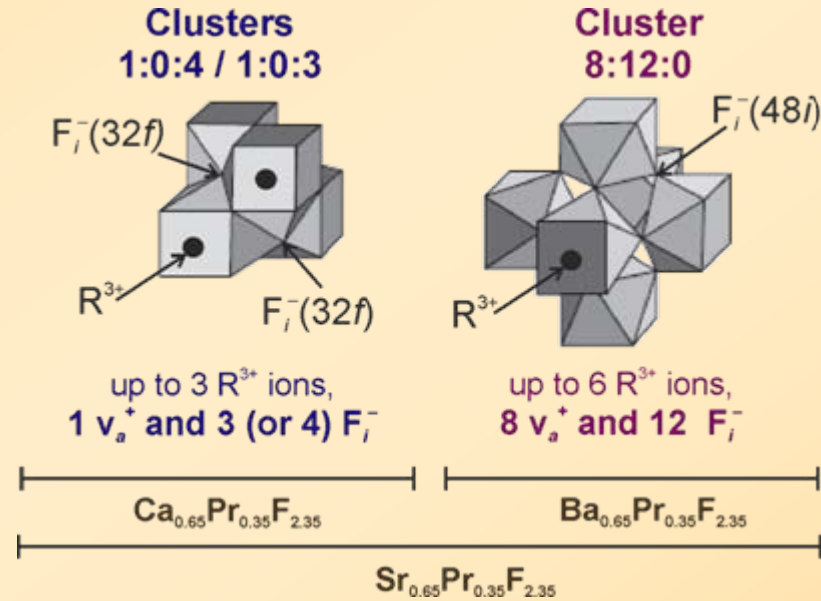
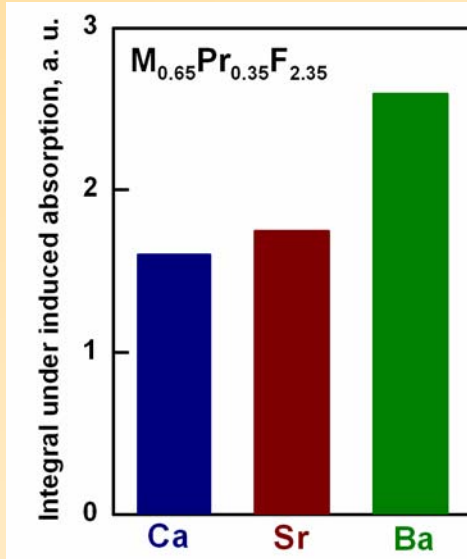
- ✓ Absorption bands typical for Pr²⁺ ions are absent

Induced absorption vs. irradiation time



- ✓ Contrary to MF₂:Pr³⁺ induced absorption of mixed crystals shows linear dependence on irradiation time without saturation region → color centers formation in M_{0.65}Pr_{0.35}F_{2.35} due to pre-irradiation defects

Coloration efficiency, D=80 Gy



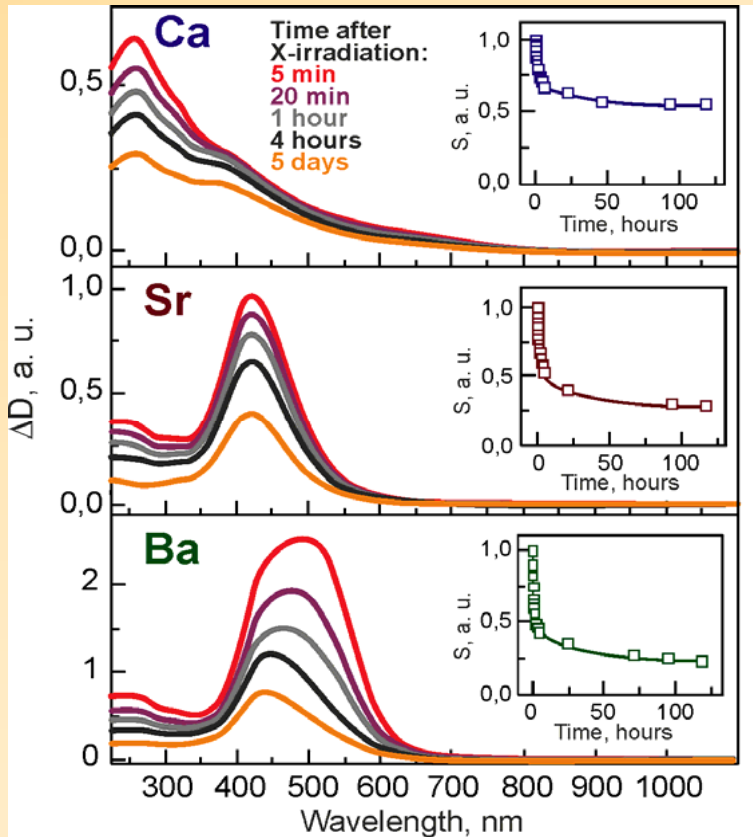
✓ the number of anion sublattice defects (V_a^+ and F_i^-) is higher for 8:0:12 clusters than for 1:0:3/1:0:4 ones

Increase in coloration efficiency in the row Ca→Sr→Ba is in agreement with cluster formation model

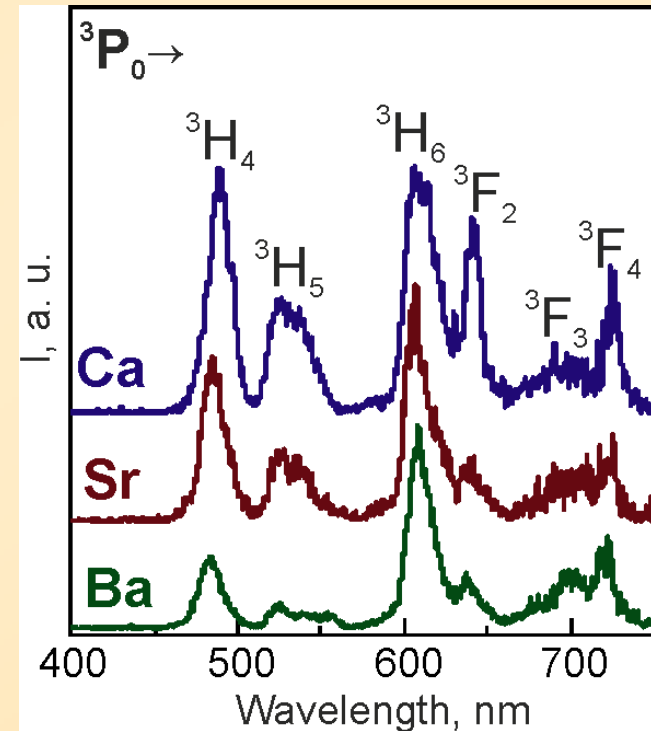
Color centers destruction

X-irradiation, 300 K, D=80 Gy

Induced absorption



Afterglow

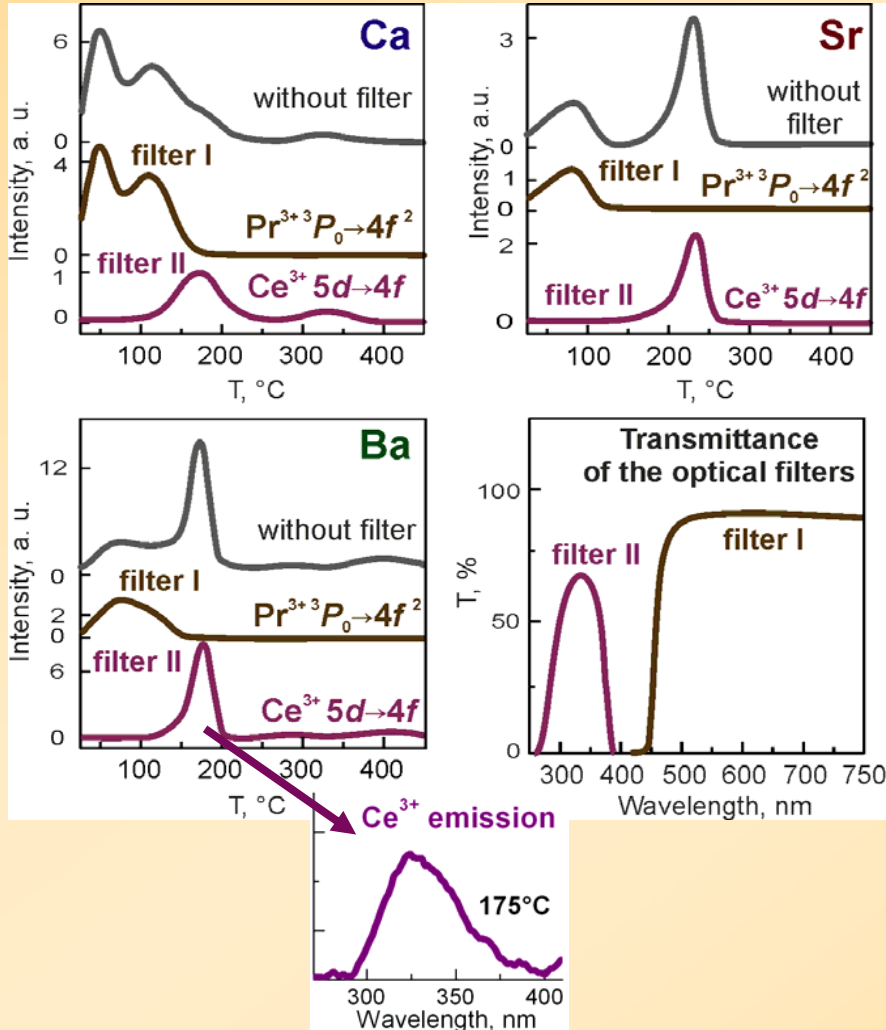


✓ Color centers destruction shows two stages → the presence of defects with different thermal stability

✓ the first stage of color centers destruction is accompanied by the afterglow corresponded $^3P_0 \rightarrow 4f^2$ radiative transitions in Pr^{3+}

X-irradiation, 300 K, D=200 Gy

TL glow curves



✓ $5d \ 14f^1 \rightarrow 4f^2$ and $1S_0 \rightarrow 4f^2$ radiative transitions of Pr^{3+} are not observed in TSL

✓ low temperature TL peaks (50-120°C) correspond to $3P_0 \rightarrow 4f^2$ transitions in Pr^{3+}

✓ weak intensity of low temperature TL peaks:
 * thermal quenching of emission from $3P_0$ level
 * cross-relaxation between adjacent Pr^{3+} ions

✓ high temperature peaks (170-240°C) are due to $5d \rightarrow 4f$ emission of impurity Ce^{3+} ions

✓ the energy of recombination of thermally released charge carriers may be non-radiatively transferred to the nearest Pr^{3+} ion, populating the low-lying $3P_0$ -level and giving rise to correspondent emission

In general

- ✓ Crystals of the $M_{1-x}R_xF_{2+x}$ solid solutions grown from melt are stable materials (no signs of decomposition during storage and thermal cycling).
- ✓ Mixed crystals open the potential possibility for modification of fluoride material properties via the selection of a proper composition.

In particular

- ✓ Various emission centers observed in $M_{1-x}Pr_xF_{2+x}$ ($M^{2+}=Ca, Sr, Ba, x=0.35$) solid solutions may be assigned to Pr^{3+} ions in different cluster types.
- ✓ Modification of the crystal structure due to the process of clustering leads to changing in the local environment of Pr^{3+} ions, and as the result to shifting of the lowest excited $5d$ level.
- ✓ Efficiency of color centers formation for $M_{1-x}Pr_xF_{2+x}$ ($M^{2+}=Ca, Sr, Ba, x=0.35$) crystals is higher than for $MF_2:Pr$ due to high concentration of pre-irradiation defects. The growth of coloration efficiency in the row $Ca \rightarrow Sr \rightarrow Ba$ is due to the presence of the different cluster types in these systems.

Thank you for attention!